

DMS 2015 MODELES A CRITERES MULTIPLES

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Plan

- 1 Introductive remarks
- 2 Ingredient of classical elasto-(visco)plastic constitutive equations
- 3 J_2 multimechanism models
 - Formulation
 - 2M2C and 2M1C
 - Typical behavior of 2M2C and 2M1C models
 - Ratchetting for 2M2C and 2M1C
- 4 Identification of the model
- 5 Concluding remarks
- 6 Bonus : extensions of the model

Purpose of the talk

- Briefly present the thermodynamical framework for unified approaches
- Introduce multimechanism models, J_2 type
- Illustrate the capabilities of the new model with reference to a ratchetting data base on 316 Stainless Steel

State variable, hardening variables

The free energy Ψ , used as a potential, defines stress and hardening variables knowing elastic strain and state variables

- Reversible part of the model
 - Elastic strain $\tilde{\epsilon}^e$ and stress $\tilde{\sigma}$

$$\tilde{\sigma} = \rho \frac{\partial \Psi}{\partial \tilde{\epsilon}^e}$$

- Dissipative part of the model
 - Strain like variables : *State variables* α_i
 - Stress like variables : *Hardening variables* A_i

$$A_i = \rho \frac{\partial \Psi}{\partial \alpha_i}$$

Classical isotropic + kinematic hardening

$$\Psi = \Psi^e + \Psi^p$$

- State variables $\tilde{\varepsilon}^e, \tilde{\alpha}, r$
- Stress $\tilde{\sigma}$
- Hardening variables \tilde{X} and \tilde{R}

$$\begin{aligned}\Psi^e &= \frac{1}{2} \tilde{\varepsilon}^e : \tilde{\Lambda} : \tilde{\varepsilon}^e & \tilde{\sigma} &= \tilde{\Lambda} : \tilde{\varepsilon}^e \\ \Psi^p &= \frac{1}{3} C \tilde{\alpha} : \tilde{\alpha} + \frac{1}{2} b Q R^2 & \tilde{X} &= \frac{2}{3} C \tilde{\alpha} \\ && & R = b Q r\end{aligned}$$

Elastic part is then fully characterized ($\tilde{\varepsilon}^e$ is observable)

Dissipative part

- Yield function $f(\tilde{\sigma}, \tilde{X}, R)$
- Flow potential $\Omega(f)$ + normality rule

$$\dot{\tilde{\varepsilon}}^p = \frac{\partial \Omega}{\partial \tilde{\sigma}} = \frac{\partial \Omega}{\partial f} \frac{\partial f}{\partial \tilde{\sigma}} = \dot{p}n$$

- Hardening potential $\Omega_h(\tilde{\sigma}, \tilde{X}, R)$

$$\dot{\tilde{\alpha}} = -\frac{\partial \Omega_h}{\partial \tilde{X}} \quad \dot{r} = -\frac{\partial \Omega_h}{\partial R}$$

- Example 1 : linear hardening, $\tilde{\alpha} = \tilde{\varepsilon}^p \quad r = p$

$$f(\tilde{\sigma}, \tilde{X}, R) = J(\tilde{\sigma} - \tilde{X}) - R - \sigma_y \quad \Omega = \Omega_h = \frac{n+1}{K} \left\langle \frac{f}{K} \right\rangle^{n+1}$$

Notation : $\dot{p} = \frac{\partial \Omega}{\partial f} = \left\langle \frac{f}{K} \right\rangle^n \quad n = \frac{\partial f}{\partial \tilde{\sigma}}$

$$\dot{\tilde{\alpha}} = \dot{p}n \quad \dot{r} = \dot{p}$$

Generalized normality rule if $\Omega_h \equiv \Omega$

Nonlinear hardening

- Additional terms in the potential

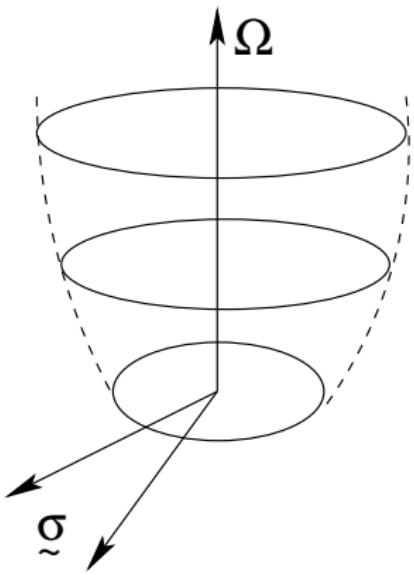
$$\Omega = \frac{n+1}{K} \left\langle \frac{f}{K} \right\rangle^{n+1} \quad \Omega_h = \Omega + \frac{3D}{4C} \tilde{X} : \tilde{X} + \frac{R^2}{2Q}$$

$$\dot{\tilde{\alpha}} = \dot{\tilde{\varepsilon}}^p - \frac{3D}{2C} \tilde{X} \dot{p} \quad \dot{r} = (1 - \frac{R}{Q}) \dot{p}$$

- Hardening variables

$$\tilde{X} = \frac{2}{3} C \dot{\tilde{\alpha}} = \frac{2}{3} C \dot{\tilde{\varepsilon}}^p - D \tilde{X} \dot{p} \quad \dot{R} = b(Q - R) \dot{p}$$

Evaluation of the dissipation



- The intrinsic dissipation is the difference between the plastic power and the power temporarily stored by the hardening mechanisms

$$\begin{aligned}\mathcal{D} &= \tilde{\sigma} : \dot{\tilde{\epsilon}}^P - \rho \dot{\Psi} \\ &= \tilde{\sigma} : \dot{\tilde{\epsilon}}^P - \tilde{X} : \dot{\tilde{\alpha}} - R \dot{p} \\ &= \tilde{\sigma} : \frac{\partial \Omega}{\partial \tilde{\sigma}} + \tilde{X} : \frac{\partial \Omega_h}{\partial \tilde{X}} + R \frac{\partial \Omega_h}{\partial R}\end{aligned}$$

\mathcal{D} is automatically positive iff $\Omega \equiv \Omega_h$ and Ω is convex

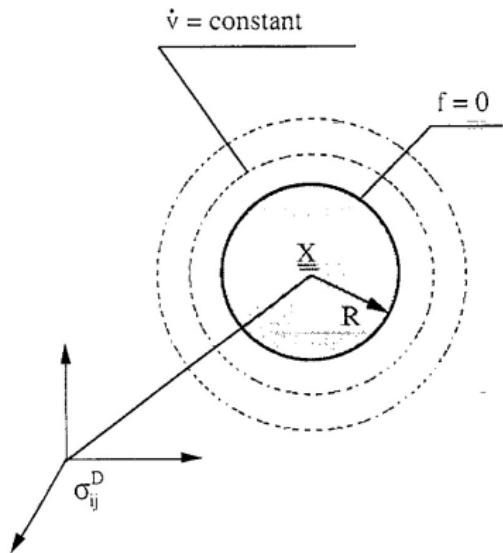
Facts concerning ratchetting

- Onedimensional ratchetting
 - Depends on hardening intensity
 - Stopped by linear hardening (either closed or open loop depending on stress range)
 - Too large with non linear hardening
 - Can be adapted by using several kinematic variables, and/or thresholds in the kinematic variables
- Twodimensional ratchetting
 - Depends on hardening intensity *and* direction
 - Too small with linear hardening
 - Too large with classical non linear hardening
 - Can be adapted by a combination of classical non linear hardening and radial fading memory

Plan

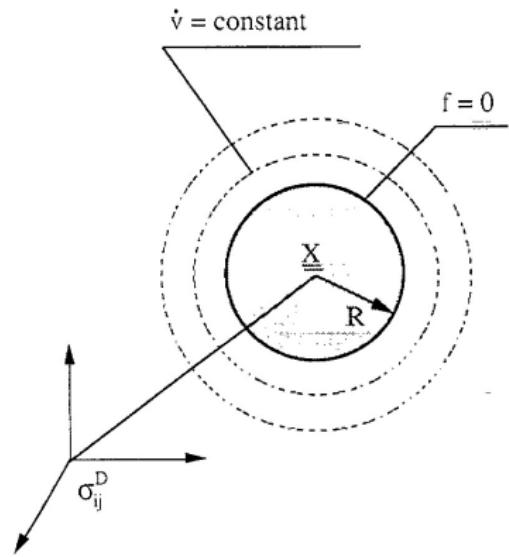
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Multimechanism *versus* unified models (1)

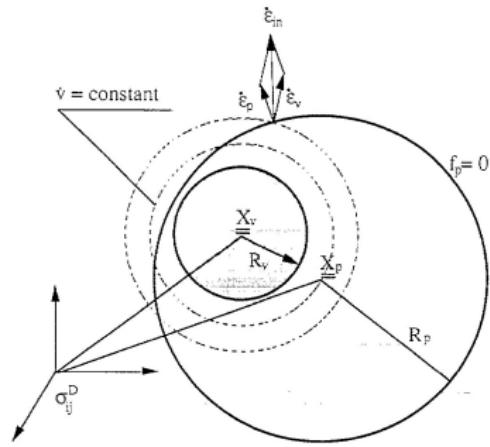


Unified model

Multimechanism versus unified models (1)



Unified model



Multimechanism

Example of a multimechanism model with von Mises local criteria

Anatomy of a multimechanism model

- Several sets of potentials, then several components of the plastic strain rate
- Several sets of $(\tilde{\sigma}^I, \tilde{X}^I, R^I, \alpha^I, r^I)$
- Stress $\tilde{\sigma}^I$ for mechanism I obtained through a concentration tensor $B^I = \frac{\partial \tilde{\sigma}^I}{\partial \tilde{\alpha}}$. *For the initial version of the models, $B^I = I$ has been chosen*
- Each $\tilde{\sigma}^I$ is involved in different yield functions $f^I(\Omega(f^I(\tilde{\sigma}^I)))$ (2M2C model and crystal plasticity)

$$\dot{\tilde{\epsilon}}^p = \sum_I \frac{\partial \Omega^I}{\partial \tilde{\sigma}} = \sum_I \frac{\partial \Omega^I}{\partial f^I} \frac{\partial f^I}{\partial \tilde{\sigma}} = \sum_I \frac{\partial \Omega^I}{\partial f^I} \tilde{n}^I : \tilde{B}^I \quad \text{several multipliers}$$

- Each $\tilde{\sigma}^I$ is involved in a global criterion $f(\Omega(f(\tilde{\sigma}^I)))$ (2M1C model)

$$\dot{\tilde{\epsilon}}^p = \frac{\partial \Omega}{\partial \tilde{\sigma}} = \frac{\partial \Omega}{\partial f} \frac{\partial f}{\partial \tilde{\sigma}} = \frac{\partial \Omega}{\partial f} \sum_I \tilde{n}^I : \tilde{B}^I \quad \text{one multiplier}$$

Each mechanism is either plastic or viscoplastic

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2M2C model : state variables and flow

- 2 mechanisms ($\alpha^1, \alpha^2, r^1, r^2$), two criteria (yield functions)
- Free energy, depends on $\alpha^1, \alpha^2, r^1, r^2$:

$$\rho\Psi = \frac{1}{3} \sum_I \sum_J C_{IJ} \tilde{\alpha}^I : \tilde{\alpha}^J + \frac{1}{2} \sum_I b_I Q_I (r^I)^2$$

$$\tilde{X}^I = \frac{2}{3} \sum_J C_{IJ} \tilde{\alpha}^J \quad R^I = b_I Q_I r^I$$

- Potential, sum of two terms :

$$f^I = J(\tilde{\sigma} - \tilde{X}^I) - R^I - R_0^I \quad \Omega = \Omega^1(f^1) + \Omega^2(f^2)$$

$$\dot{\tilde{\epsilon}}^P = \frac{\partial \Omega^1}{\partial f^1} \tilde{n}^1 + \frac{\partial \Omega^2}{\partial f^2} \tilde{n}^2 \text{ with } \tilde{n}^I = \frac{\partial f^I}{\partial \tilde{\sigma}}$$

Each flow can be viscoplastic or plastic

2M1C model : state variables and flow

- 2 mechanisms ($\tilde{\alpha}^1, \tilde{\alpha}^2, r$), one criterion (yield function)
- Free energy, depends on $\tilde{\alpha}^1, \tilde{\alpha}^2, r$:

$$\rho\psi = \frac{1}{3} \sum_I \sum_J C_{IJ} \tilde{\alpha}^I : \tilde{\alpha}^J + \frac{1}{2} bQr^2$$

$$\tilde{X}^I = \frac{2}{3} \sum_J C_{IJ} \tilde{\alpha}^J \quad R = bQr$$

- Potential, depends on one yield only :

$$f = (J(\tilde{\sigma} - \tilde{X}^1)^2 + J(\tilde{\sigma} - \tilde{X}^2)^2)^{1/2} - R - R_0 \quad \Omega \equiv \Omega(f)$$

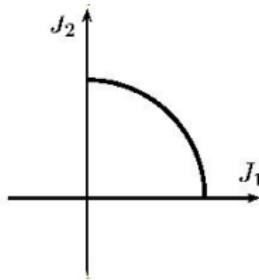
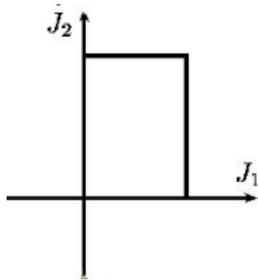
$$\dot{\tilde{\xi}}^P = \frac{\partial \Omega}{\partial f} \tilde{n} \quad \text{with} \quad \tilde{n} = \frac{J_1 \tilde{n}^1 + J_2 \tilde{n}^2}{(J_1^2 + J_2^2)^{1/2}}$$

$$\text{and } J_I = J(\tilde{\sigma} - \tilde{X}^I) \quad \tilde{n}^I = \frac{3}{2} \frac{\tilde{s} - \tilde{X}^I}{J^I}$$

Only one (visco)plastic flow

Comparison between 2M2C and 2M1C model features

- Application of the normality rule



- Coupling between the hardening variables

$$\begin{pmatrix} \tilde{X}_1 \\ \tilde{X}_2 \end{pmatrix} = \begin{pmatrix} C_{11} & C_{12} \\ C_{12} & C_{22} \end{pmatrix} \begin{pmatrix} \alpha_1 \\ \alpha_2 \end{pmatrix}$$

- Ratchetting iff the determinant $C_{11}C_{22} - C_{12}^2 = 0$*

2M2C model : hardening

- Hardening rules :

$$\dot{\tilde{\alpha}}^I = \left(\tilde{n}^I - \frac{3D_I}{2C_{II}} \tilde{X}^I \right) \frac{\partial \Omega^I}{\partial f^I} \quad \dot{r} = \left(1 - \frac{R^I}{Q_I} \right) \frac{\partial \Omega^I}{\partial f^I}$$

- Denote $\dot{p}^I = \frac{\partial \Omega^I}{\partial f^I}$ for plastic, and $\dot{v}^I = \frac{\partial \Omega^I}{\partial f^I}$ for viscoplastic flow
- Three possible models :

2M2C-VV, two viscoplastic strains

$$\dot{\tilde{\epsilon}}^P = \dot{v}^1 \tilde{n}^1 + \dot{v}^2 \tilde{n}^2$$

2M2C-PP, two plastic strains

$$\dot{\tilde{\epsilon}}^P = \dot{p}^1 \tilde{n}^1 + \dot{p}^2 \tilde{n}^2$$

2M2C-VP, one plastic, one viscoplastic

$$\dot{\tilde{\epsilon}}^P = \dot{v}^1 \tilde{n}^1 + \dot{p}^2 \tilde{n}^2$$

2M2C-PP model

For two mechanisms :

$$\dot{\lambda}^1 = \left\langle \frac{M_{22}\tilde{n}^1 : \dot{\sigma} - M_{12}\tilde{n}^2 : \dot{\sigma}}{M_{11}M_{22} - M_{12}M_{21}} \right\rangle$$

$$\dot{\lambda}^2 = \left\langle \frac{M_{11}\tilde{n}^2 : \dot{\sigma} - M_{21}\tilde{n}^1 : \dot{\sigma}}{M_{11}M_{22} - M_{12}M_{21}} \right\rangle$$

with

$$M_{II} = C_{II} - D_I \tilde{X}^I : \tilde{n}^I + b_I(Q_I - R^I)$$

$$M_{IJ} = \frac{2}{3} C_{IJ} \tilde{n}^I : \tilde{n}^J - D_J \frac{C_{IJ}}{C_{JJ}} \tilde{n}^I : \tilde{X}^J$$

2M2C-VP model (1)

Viscoplastic part ($1 \equiv v$)

$$\begin{aligned} f^v &= J(\tilde{\sigma} - \tilde{X}^v) - R^v - R_{ov} \\ \tilde{X}^v &= (2/3)C_v\tilde{\alpha}^v + C_{vp}\tilde{\alpha}^p \\ R^v &= b_v Q_v r^v \end{aligned}$$

$$\dot{\tilde{\alpha}}^v = \dot{\tilde{\varepsilon}}^v - \left(\frac{3D_v}{2C_v} \right) \tilde{X}^v \dot{v}$$

$$\dot{r}^v = \left(1 - \frac{R^v}{Q_v} \right) \dot{v}$$

$$\dot{v} = \left\langle \frac{f^v}{K} \right\rangle^n$$

Plastic part ($2 \equiv p$)

$$\begin{aligned} f^p &= J(\tilde{\sigma} - \tilde{X}^p) - R^p - R_{op} \\ \tilde{X}^p &= (2/3)C_p\tilde{\alpha}^p + C_{vp}\tilde{\alpha}^v \\ R^p &= b_p Q_p r^p \end{aligned}$$

$$\dot{\tilde{\alpha}}^p = \dot{\tilde{\varepsilon}}^p - \left(\frac{3D_p}{2C_p} \right) \tilde{X}^p \dot{\lambda}$$

$$\dot{r}^p = \left(1 - \frac{R^p}{Q_p} \right) \dot{p}$$

$$\dot{\lambda} = \frac{<\dot{\tilde{\sigma}} - C_{vp}\dot{\tilde{\alpha}}^v> : n^p}{H_p}$$

$$\text{with } H_p = C_p - D_p \tilde{X}^p : n^p + b_p (Q_p - R^p)$$

2M2C-VP model (2)

Remark : consistency condition for 2M2C-VP model

$$\dot{f}^p = \tilde{n}^p : \dot{\boldsymbol{\sigma}} - \tilde{n}^p \tilde{X}^p - \dot{R}^p$$

Coupling through :

$$\tilde{X}^p = C_{pp} \dot{\tilde{\alpha}}^{pp} + C_{vp} \dot{\tilde{\alpha}}^{vp}$$

$$\tilde{X}^p = C_{pp} \left(\tilde{n}^p - \frac{3D_p}{2C_{pp}} \tilde{X}^p \right) \dot{p} + C_{vp} \left(\tilde{n}^v - \frac{3D_v}{2C_{vv}} \tilde{X}^v \right) \dot{v}$$

$$\dot{R}^p = b_p Q_p \left(1 - \frac{R^p}{Q_p} \right) \dot{p}$$

The plastic increment \dot{p} is now time dependent...

2M2C-VP model(3)

- Limitation of the viscous stress for very high strain rate by the time independent model
- The yield limit for creep can be much lower than yield limit for inviscid plasticity
- Full/limited coupling between plasticity and creep
- Special coefficient sets provide *inverse strain rate* effect (lower stress for higher strain rate)

2M1C-P : plastic multiplier

$$\dot{\lambda} = \left\langle \frac{(J_1 \tilde{n}^1 + J_2 \tilde{n}^2) : \dot{\tilde{\sigma}}}{h_R + h_{X1} + h_{X2}} \right\rangle$$

with :

$$h_R = b(Q - R)R$$

$$h_{X1} = \frac{2}{3} \left(\tilde{n}^1 - \frac{3D_1}{2C_{11}} \tilde{X}^1 \right) : (C_{11}J_1 \tilde{n}^1 + C_{12}J_2 \tilde{n}^2)$$

$$h_{X2} = \frac{2}{3} \left(\tilde{n}^2 - \frac{3D_2}{2C_{22}} \tilde{X}^2 \right) : (C_{12}J_1 \tilde{n}^1 + C_{22}J_2 \tilde{n}^2)$$

No ratchetting, except if $C_{11}C_{22} - C_{12}^2 = 0$

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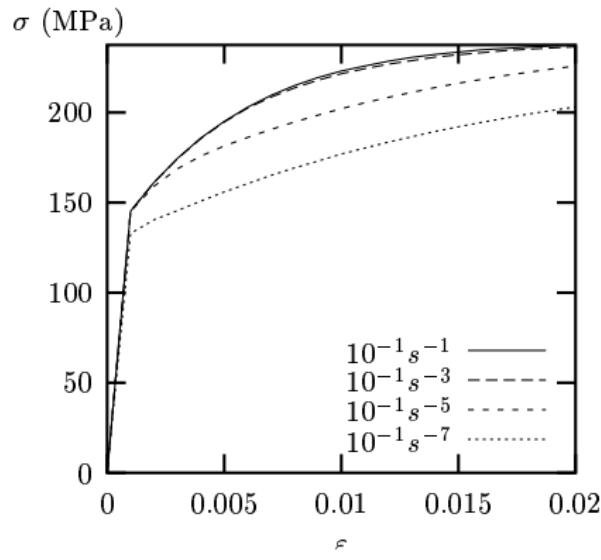
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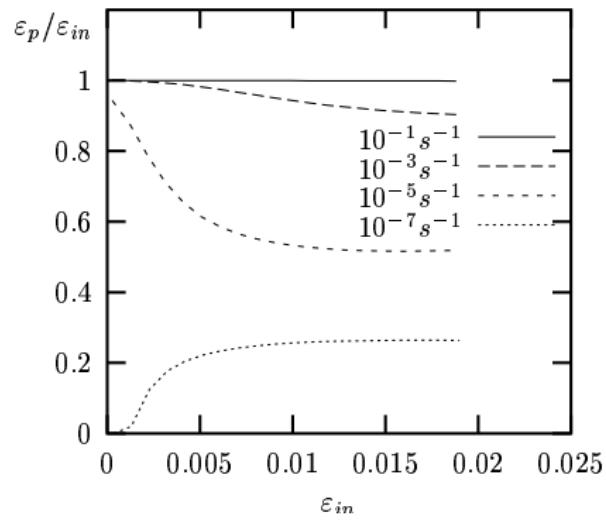
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2M2C-VP model : Balance between plastic and viscoplastic strain



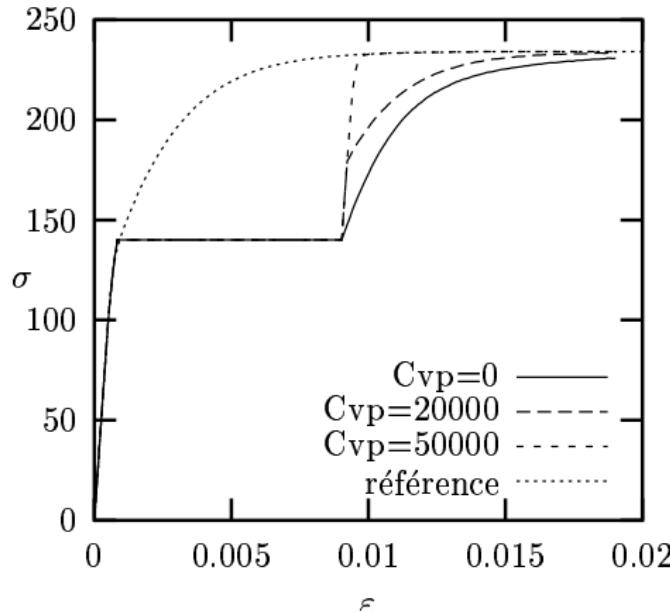
Tensile curves at various strain rates



Amount of plasticity in inelastic strain

$$K = 500; n = 7; R_0^V = 80; C_V = 10000; D = 100; R_0^P = 140; C_P = 20000; D = 200$$

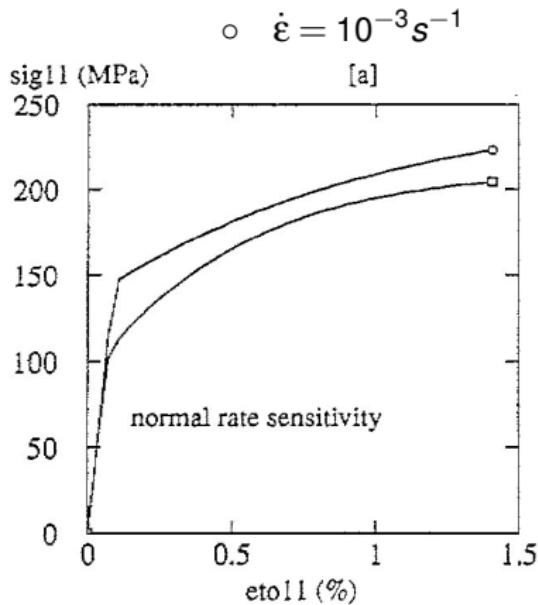
2M2C-VP model : Influence of the coupling term



(units : MPa, s)
viscosity :
$K = 500 ; n = 7$
isotropic viscous :
$R_0 = 0$
kinematic viscous :
$C = 50000 ; D = 500$
isotropic plastic :
$R_0 = 140$
kinematic plastic :
$C = 50000 ; D = 500$

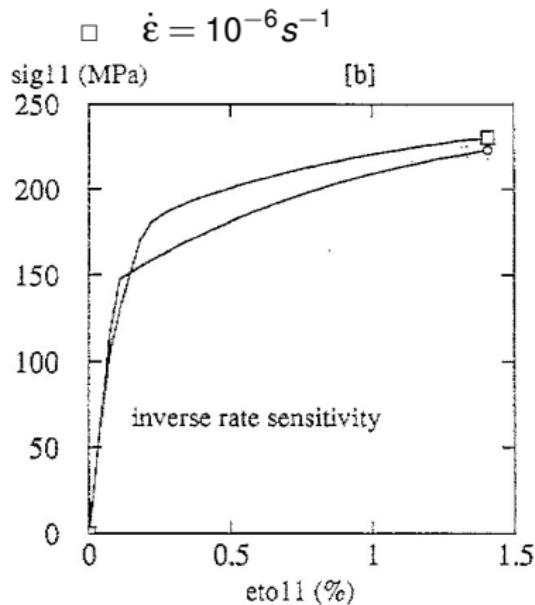
Creep 555 h at 140 MPa, then tension at $\dot{\epsilon} = 10^{-4} \text{ s}^{-1}$
reference in tension at $\dot{\epsilon} = 10^{-4} \text{ s}^{-1}$

2M2C-VP model : Inverse strain rate effect



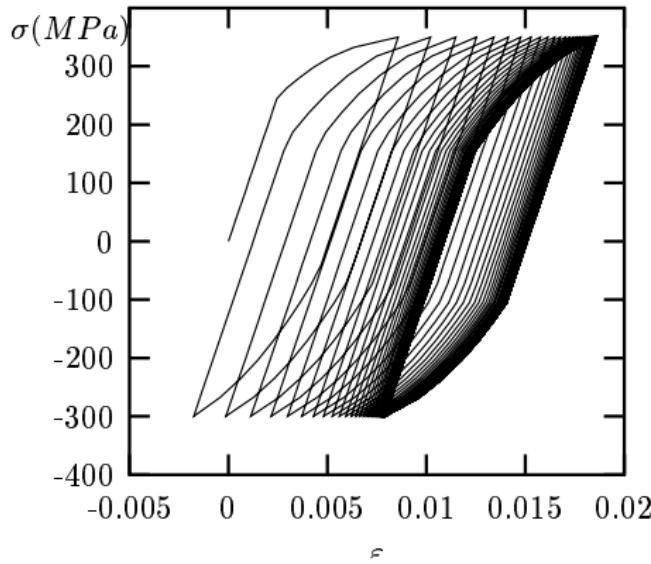
Normal rate sensitivity if $C_p < C_v < C_{vp}$
 $C_p = 10000 < C_v = 20000 < C_{vp} = 40000$
 $D_p = 100, D_v = 200$

coefficients : $K = 700, n = 7.2; R_0^V = 0; R_0^P = 140$



Inverse rate sensitivity if $C_p < C_{vp} \ll C_v$
 $C_p = 10000 < C_{vp} = 40000 < C_v = 100000$ Other
 $D_p = 400, D_v = 100$

2M1C-P : Plastic shakedown



coefficients (MPa,s)

viscosité :

n=2, K=5

cinématique I :

$C_{11} = 2000$; $D_1 = 0$

cinématique II :

$C_{22} = 200000$; $D_2 = 0$

isotrope :

$R_0 = 350$

couplage :

$C_{12} = -20000$

Steady state presenting an open loop under non symmetrical loading

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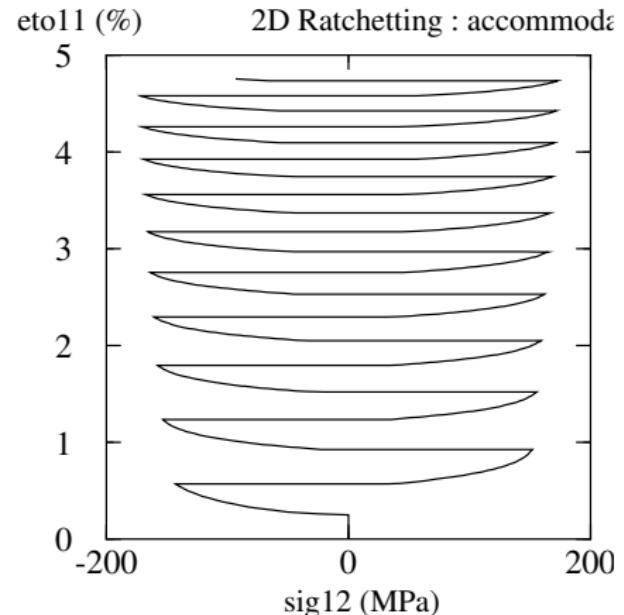
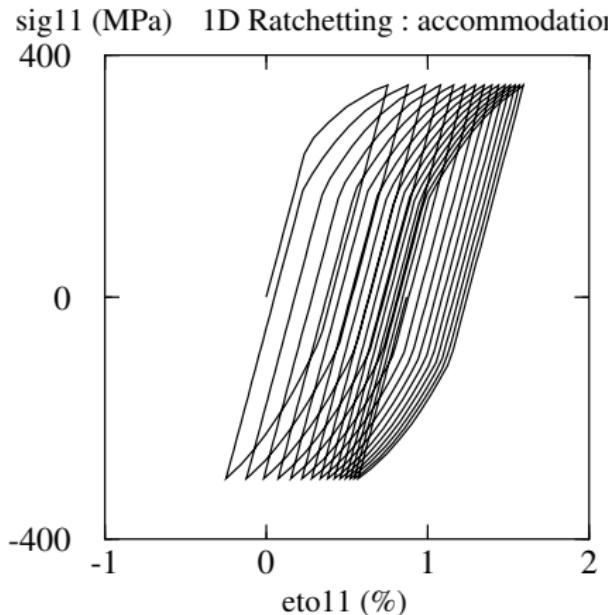
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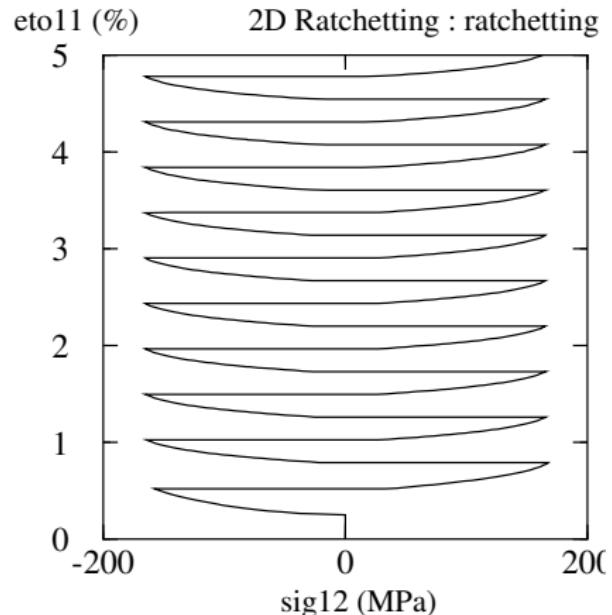
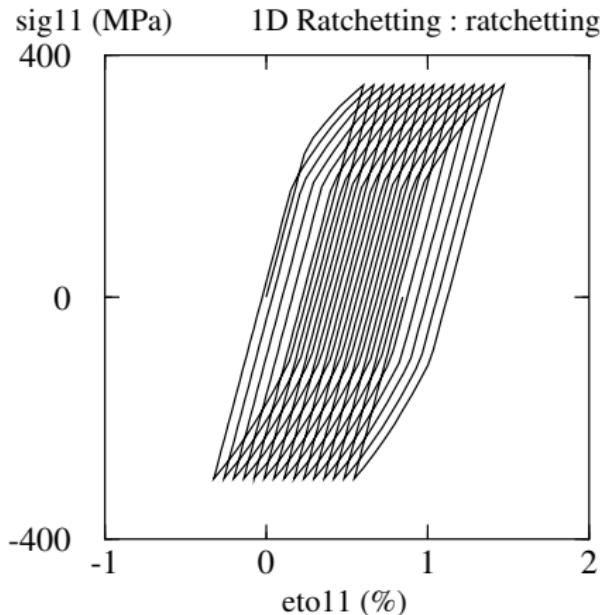
6 Bonus : extensions of the model

Ratchetting behavior, Regular Matrix



No ratchetting since the determinant $C_{11} C_{22} - C_{12}^2 \neq 0$

Ratchetting behavior, Singular Matrix



Ratchetting since the determinant $C_{11} C_{22} - C_{12}^2 = 0$

Possible improvements

- The ratchetting behavior in the multimechanism models is the result of :
 - The character of the hardening matrix (Regular or Singular)
 - The evolution rules of the kinematic hardening variables (Linear or Non Linear)
- The source of improvements are :
 - The localization rules of the micro-mechanical (β -rule)
 - The evolution rules established for the unified models

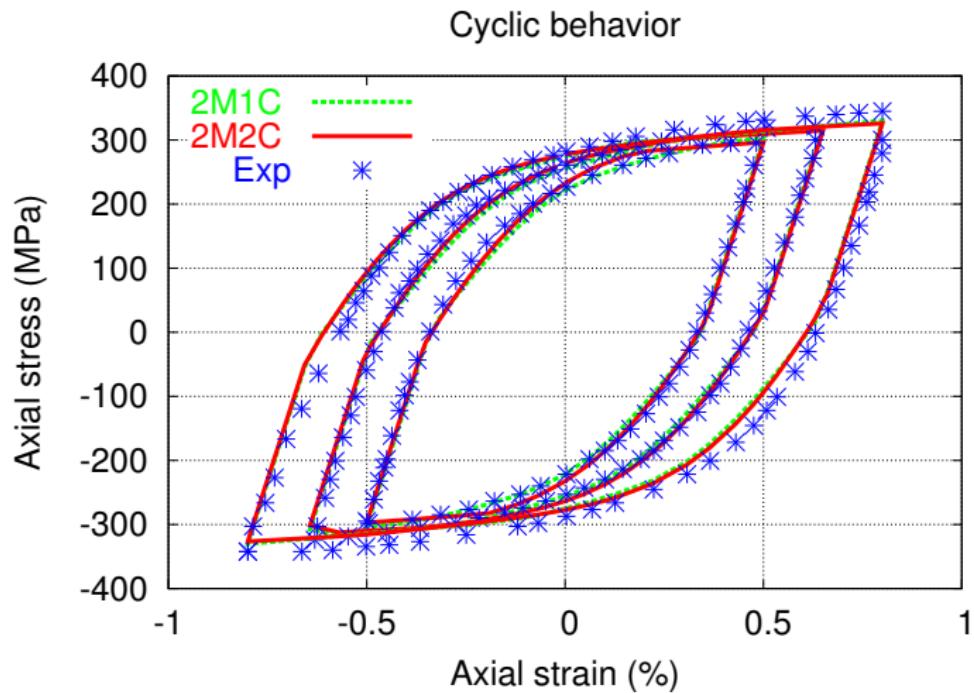
An identification of the model

Experimental data base of a 316 stainless steel ([Portier et al., 2000])
tests at room temperature (25°C)

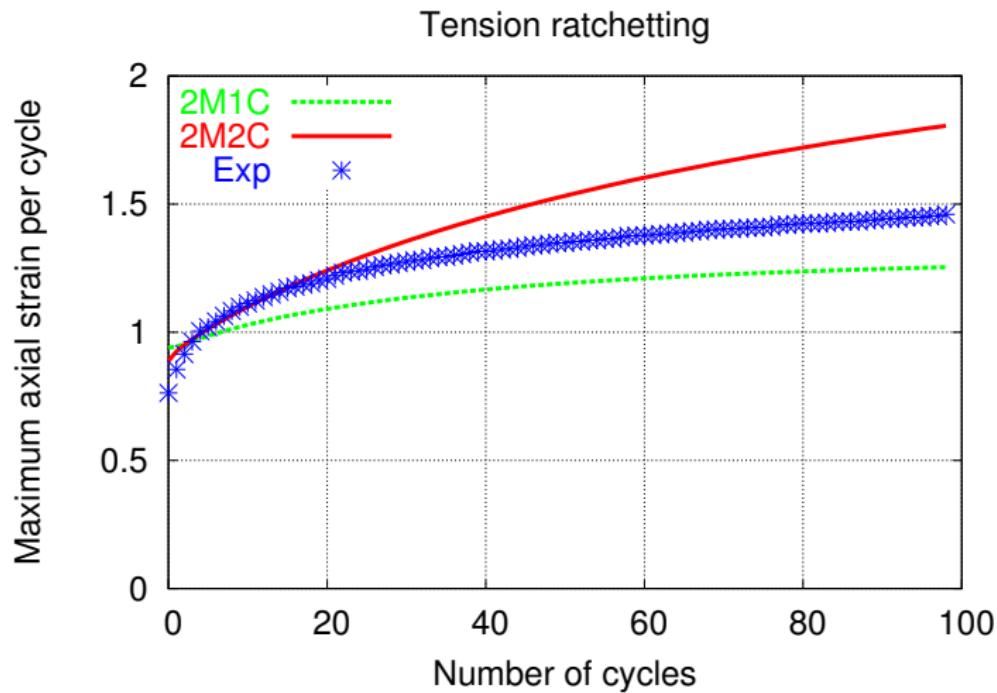
- Monotonic tensile tests,
- Cyclic uni-axial tension-compression for three strain ranges,
- Tension-torsion ratchetting tests with two values of tensile stress and with different shear strain amplitude,
- Tension-torsion out-of-phase test in the steady-state stress response.

$2\text{M1C}_{-\beta}$ and $2\text{M2C}_{-\beta}$ have been identified

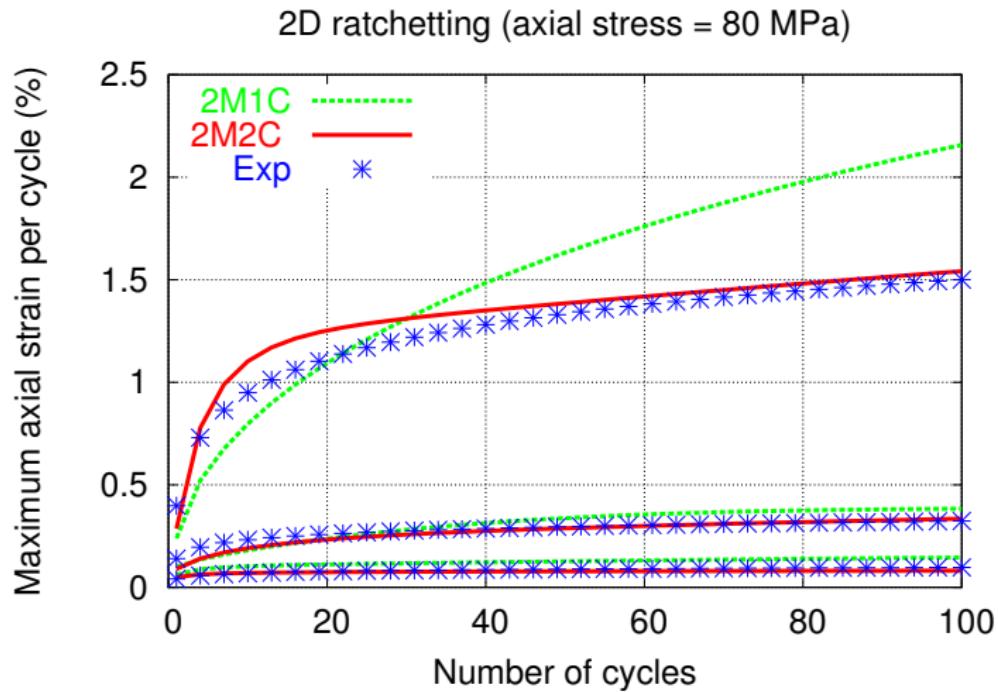
Cyclic tests



Onedimensional ratchetting



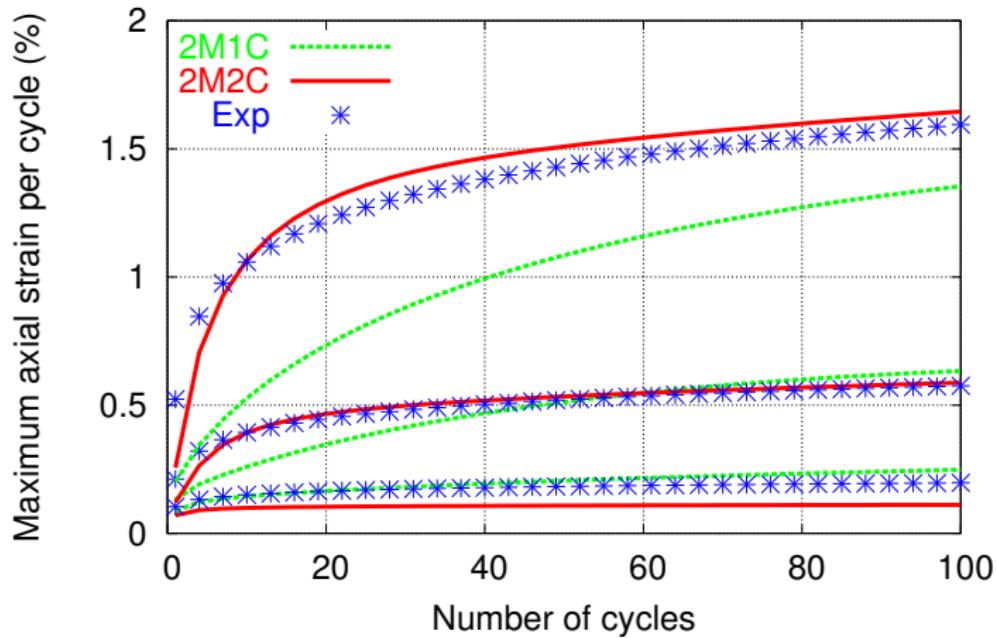
Twodimensional ratchetting (1)



Axial stress 80 MPa

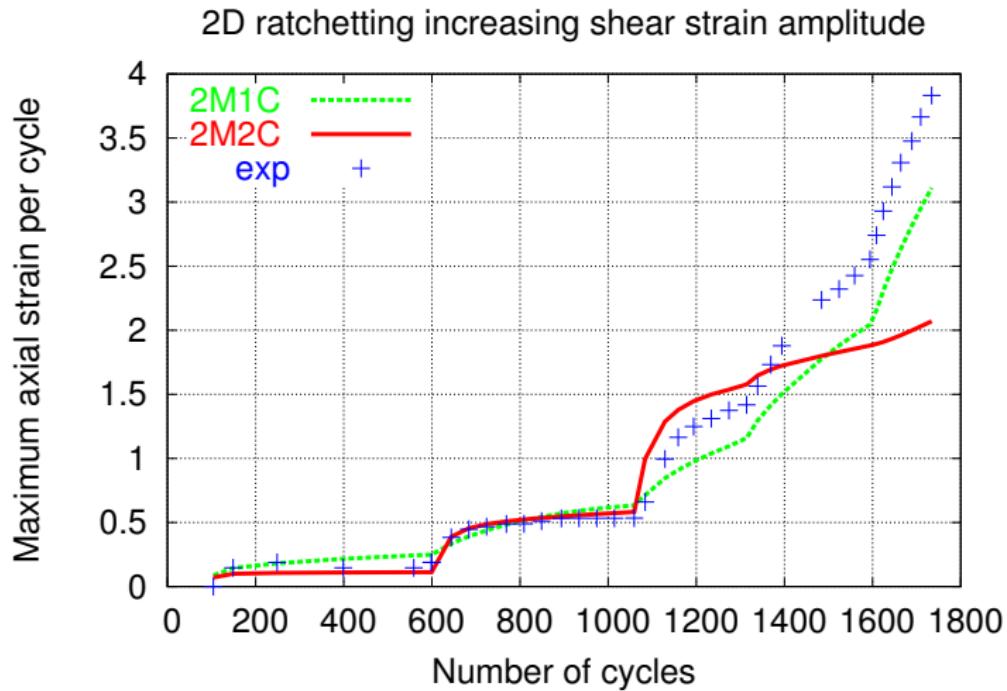
Twodimensional ratchetting (2)

2D ratchetting (axial stress = 100 MPa)

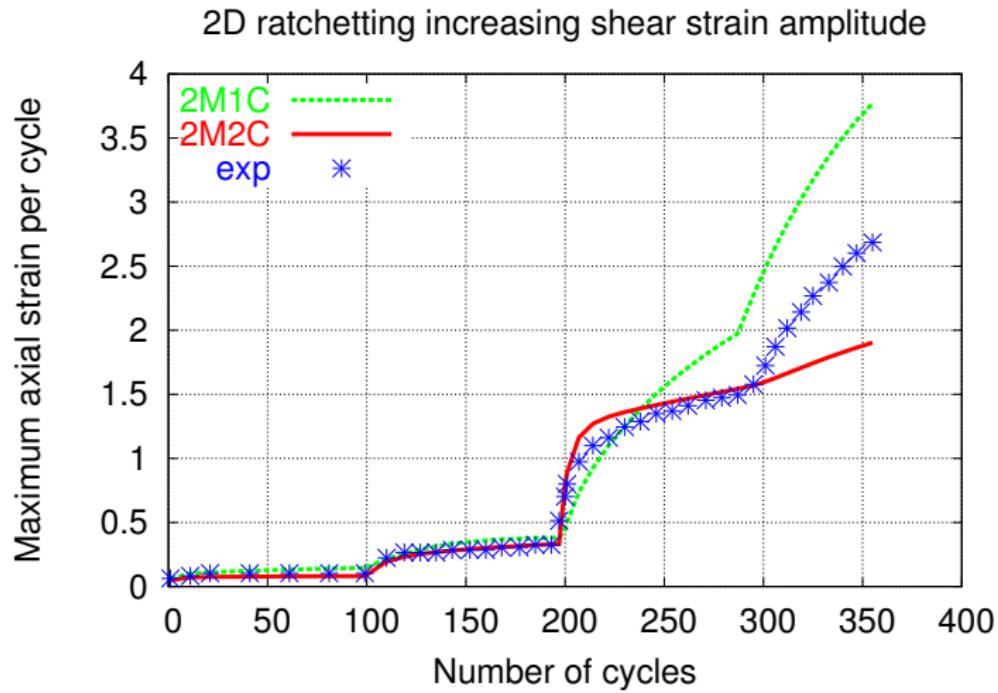


Axial stress 100 MPa

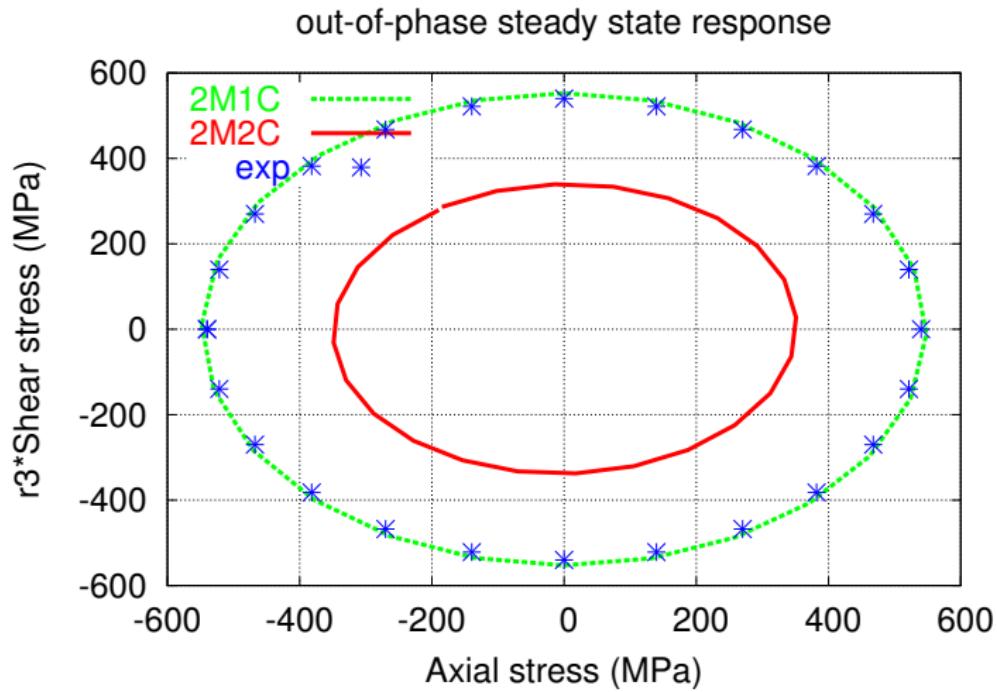
2D ratchetting, increasing strain amplitude (1)



2D ratchetting, increasing strain amplitude (2)



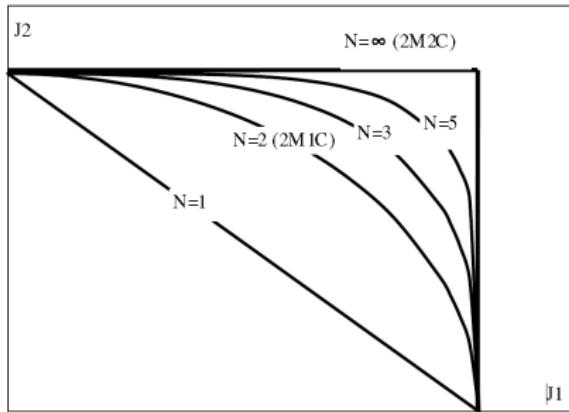
Out-of-phase test



Concluding remarks

- Other versions are possible in the $nMmC$ model class
- Fully equipped with singular and regular/singular matrices
- Choices of fading memory terms, strain memory effect, etc.
- High versatility *wrt* ratchetting behavior, additional hardening
- Several groups are involved in new developments
 - K. Saï (ENIS Sfax, Tunisia) [[Cailletaud and Saï, 1995](#), [Sai and Cailletaud, 2006](#),
[Saï and Cailletaud, 2007](#), [Cailletaud and Saï, 2008](#), [Saï et al., 2012](#), [Saï et al., 2014](#)]
 - L. Taleb (INSA Rouen, France) [[Taleb et al., 2006](#), [Taleb and Cailletaud, 2010](#),
[Taleb and Cailletaud, 2011](#), [Saï et al., 2014](#)]
 - M. Wolff (Univ. Bremen, Germany) [[Wolff and Taleb, 2008](#)]

Alternative form of 2M1C's criterion



- Instead of

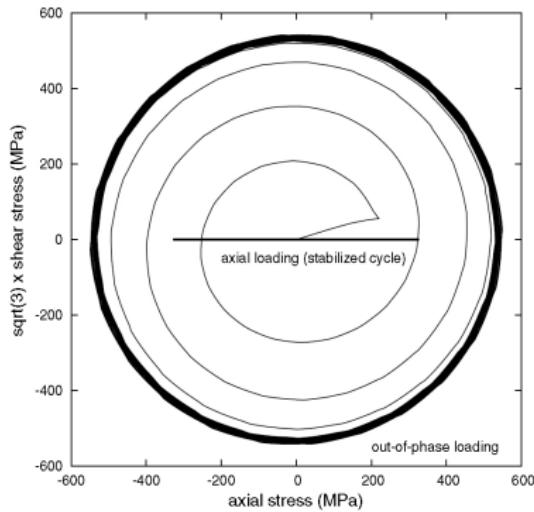
$$f = (J(\tilde{\sigma} - \tilde{X}^1)^2 + J(\tilde{\sigma} - \tilde{X}^2)^2)^{1/2} - R - R_0 = (J_1^2 + J_2^2)^{1/2} - R - R_0$$

- Write

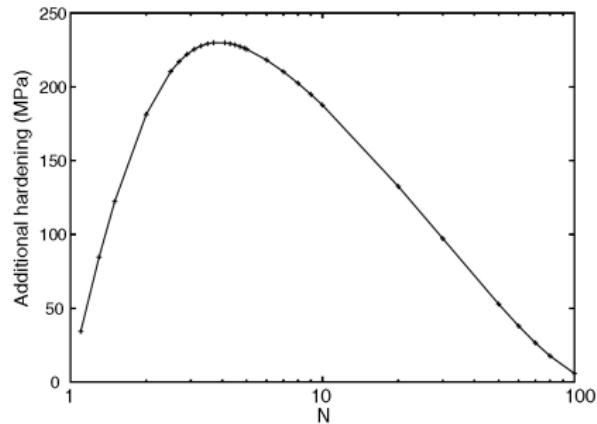
$$f = (J(\tilde{\sigma} - \tilde{X}^1)^N + J(\tilde{\sigma} - \tilde{X}^2)^N)^{1/N} - R - R_0 = (J_1^N + J_2^N)^{1/N} - R - R_0$$

- Continuous transition from 2M1C ($N = 2$) to 2M1C ($N \rightarrow \infty$)

Influence of N on additional hardening



Stress response for 90° phase lag
in strain space



Additional hardening
versus N

Maximum for $N = 3$, optimum wrt experiment, $N = 2$ (or $N = 4$)

Alternative form of the fading memory term

- Instead of using \tilde{X}^I ,

$$\dot{\tilde{\alpha}}^I = \left(\tilde{n}^I - \frac{3D_I}{2C_{II}} \tilde{X}^I \right) \dot{p}^I$$

1/ Use $\tilde{\alpha}^I$

$$\dot{\tilde{\alpha}}^I = (\tilde{n}^I - D_I \tilde{\alpha}^I) \dot{p}^I$$

(the thermodynamical aspect is then more difficult to achieve)

2/ Use $(1 - \eta)\alpha^I + \eta(\alpha^I : \tilde{n}^I)\tilde{n}^I$

$$\dot{\tilde{\alpha}}^I = (\tilde{n}^I - D_I [(1 - \eta)\alpha^I + \eta(\alpha^I : \tilde{n}^I)]) \dot{p}^I$$

(one more parameter, η , to adjust the direction of the fading memory term)

Warning : thermodynamics must be respected

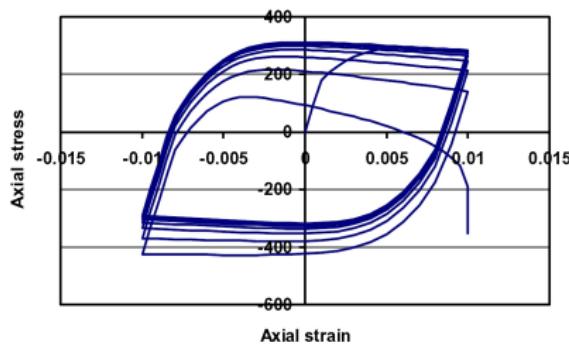
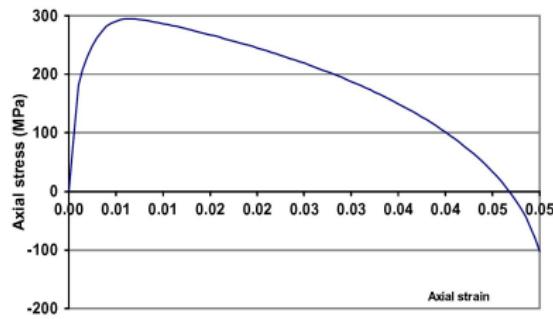
- General condition

$$C_{11} C_{22} - C_{12}^2 \geq 0$$

- In case of a fading memory with η

$$\eta \leq 1 - \sqrt{\frac{C_{12}^2(D_1 + D_2)^2}{4C_{11}C_{22}D_1D_2}}$$

- Example with $C_{11} C_{22} - C_{12}^2 < 0$



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